

# 알루미늄 5052-O 합금과 연강의 맞대기 마찰교반접합에 관한 연구

## Friction Stir Dissimilar Butt Welding of Mild Steel and Aluminum 5052-O Alloy

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*The solid state dissimilar joining of mild steel and aluminum 5052-O alloy is successfully achieved by friction stir welding (FSW). The 2 mm thick sheets are butt welded using a convex scrolled tool made of tungsten carbide. With a constant weld speed of 75 mm/min, two different tool at rotation speeds of, 800 and 1000 rpm, were employed to determine the feasibility of the joint formation. Macroscopic observation of the cross section confirmed the formation of a sound FSW joint. However, the formation of an intermetallic in the Stir Zone (SZ) is also observed for the both sets of process parameters. Comparatively, better material mixing is observed when the parameters are set at, 1000 rpm and 75 mm/min respectively. The hardness test revealed the presence of three distinct hardness zones in the SZ for the two parameter sets.*

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### 1. Introduction

The environmental and energy efficiency concerns in transportation industries compel manufacture of multi-material and hybrid structures. These new structures frequently use lightweight alloys like aluminum or magnesium alloys in their parts with the aim of overall weight reduction for lowering fuel consumptions and toxic emission. Engineering relevance involves joining of aluminum alloy and steels as an important dissimilar fabrication requirement for the manufacture of multi-material structures.<sup>1,2</sup> Due to the significantly different melting temperatures and other physical properties, joining of aluminum alloy and steels by conventional fusion joining process is extremely difficult.

A conventional fusion joining process like arc welding leads to formation of complex weld pool shapes, inhomogeneous

solidification microstructures, and segregations. Also the extremely low solubility of Fe in Al leads to the formation of brittle and excessive Al-rich  $Fe_xAl_y$  intermetallic compound (IMC) phases, which are detrimental to the mechanical properties of the joint.<sup>3</sup> In this regard, Kreimeyer and Sepold<sup>4</sup> suggested that if the IMC layer is less than 10  $\mu m$  thick, the joint may be mechanically sound.

Various joining methods such as friction stir knead welding,<sup>5</sup> friction welding,<sup>6,7</sup> surface activated bonding,<sup>8</sup> abrasion circle friction spot welding,<sup>9</sup> cold metal transfer,<sup>10</sup> laser penetration welding,<sup>11</sup> and barrel nitriding process<sup>12</sup> have been used to join aluminum alloys to steels. Among the mentioned joining methods, friction stir welding (FSW) does not require a high pressure or vacuum chamber. Therefore, FSW can be easily applied inside the assembly plant for both continuous and spot welding processes.<sup>13</sup> The joining mechanism in butt FSW mainly involves intermixing

of plastically deformed materials in stir zone (SZ). Inside the SZ, high temperature and strain rate generally lead to the formation of dynamically recrystallized fine grain structure, which consequently enhances the mechanical properties of the joint.<sup>14</sup> Even though initial investigations on FSW of aluminum alloys to steels report the presence of IMCs such as  $Fe_4Al_{13}$ ,  $Fe_2Al_5$ , and  $FeAl_4$ , the formation of IMC layers at aluminum-steel interface weakens significantly due to relatively lower heat input in FSW.<sup>15</sup>

The FSW process parameters such as tool rotation speed and weld speed influence the microstructure and properties of the joint. Therefore, it is natural that the optimization of the process parameters has been the focus of the researchers. Butt FSW of thin sheets of aluminum 6061-T6 alloy and advanced high strength steel was conducted by Liu et al.<sup>16</sup> They analyzed the effect of process parameters on the joint microstructure evolution based on the mechanical welding force and temperature, which were measured during the welding process. Also, Coelho et al.<sup>17</sup> investigated the influence of a distinct high strength steel base material on the joint efficiency of FSW of aluminum alloy to the high strength steel. They concluded that the joint efficiency depended foremost on the mechanical properties of the heat affected zone (HAZ) and the thermo-mechanical affected zone (TMAZ) of the aluminum alloy.<sup>17</sup>

In this technical report, FSW of aluminum 5052-O alloy and mild steel sheets has been conducted by altering the process parameters in butt configuration with offset of the tool pin towards the steel sheet at the advancing side. The cross section of joint was briefly analyzed with the aid of optical microscopy and microhardness indentation test.

## 2. Experimental Set-Up

Mild steel (at the advancing side) and aluminum 5052-O alloy (at the retreating side) sheets of 2 mm thickness each were butt welded in displacement controlled mode using custom-made FSW machine (RM1A, MTI, USA), as schematically shown in Fig. 1. The process parameters for FSW are listed in Table 1. Note that the welding was carried out by offsetting the tool to the distance of 0.5 mm towards the steel sheet side at the advancing side. It has been reported that offsetting the tool to an aluminum alloy side serves as a better method to protect the tool from wearing during FSW of steels and aluminum alloys. However, in that case, researchers also reported no trace of material intermixing inside the SZ.<sup>18,19</sup> Material intermixing being a crucial factor for determining joint strength was found to be better in this present study when the tool was inserted into the steel side.

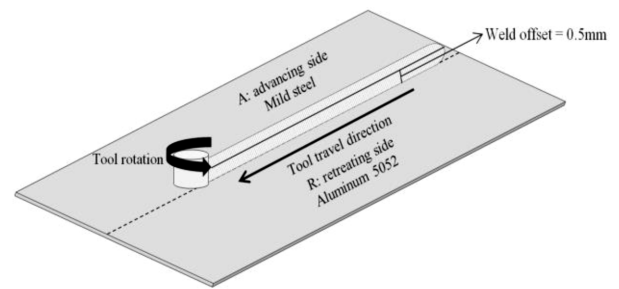


Fig. 1 Schematic representation of experiment

Table 1 Process parameters and Energy input

| Tool rotation speed (rpm) | Tool travel speed (mm/min) | Tool tilt angle (°) | Depth of penetration (mm) | Energy input (KJ) |
|---------------------------|----------------------------|---------------------|---------------------------|-------------------|
| 1000                      | 75                         | 2                   | 1.9                       | 158.7             |
| 800                       |                            |                     |                           | 123.6             |

The axial force and torque histories recorded in the data acquisition system attached to the FSW machine were first analyzed to evaluate the heat input during the process. At next, to observe the material flow in the SZ, the cross section perpendicular to the welding direction was prepared by polishing and etching with Keller's (for aluminum) and Nital (for mild steel) etchant. Optical microscopy (OM; A1m Axio Imager, Carl Zeiss, Germany) was then carried on the cross section. For the evaluation of mechanical properties of the joint, Vickers hardness was measured along the cross section using a fully calibrated Vickers Microhardness tester (A-1170, Leica, Germany) with a load of 0.49 N for 10 sec.

## 3. Results and Discussion

The mechanical interactions between the FSW tool and the workpiece materials can be easily examined from the process responses, the axial force and torque histories in Figs. 2(a) and 2(b). These process responses are meaningful since they are related with the heat input during joining by the stirring action of the pin and shoulder of the tool. The force and torques histories recorded during the FSW exhibit that by increasing the tool rotation speed from 800 to 1000 rpm at the constant weld speed of 75 mm/min, both the axial force and spindle torque decreases significantly. This clearly reveals that the frictional heat increased immensely with the increase in tool rotation speed at the constant weld speed, which directly influences plasticization of the materials. From the process responses in Fig. 2, the heat input during FSW can be approximated using the relation below.<sup>20</sup>

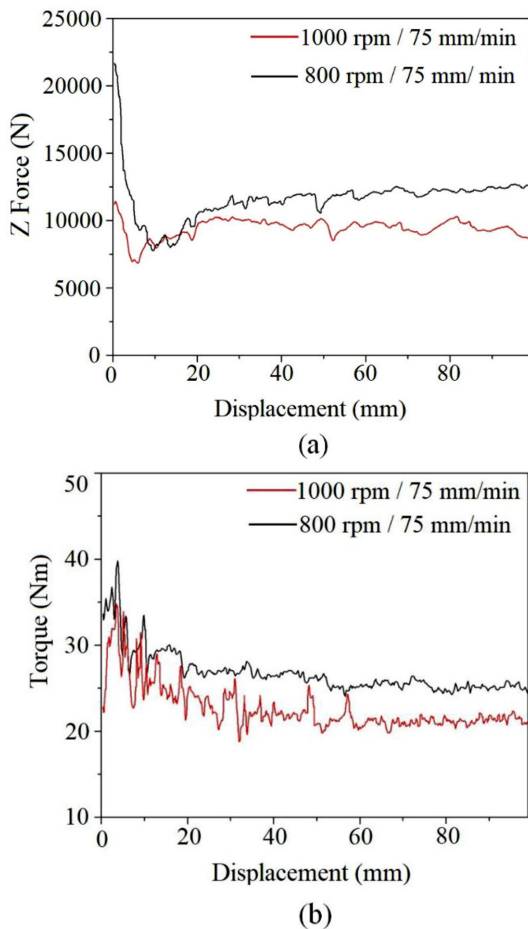


Fig. 2 Process responses: (a) Z-force and (b) torque

$$\text{Heat input} = \int_0^t C_z(t) N_p \times \frac{2\pi}{60} dt \quad (1)$$

Here,  $C_z$  and  $N_p$  represent the torque and the tool rotation speed (in rpm), respectively. The heat (or energy) input values calculated using Eq. (1) are listed in Table 1.

The material flow in the FSW joint was examined for the both FSW parameter combinations, 800 rpm - 75 mm/min and 1000 rpm - 75 mm/min, using a differential etching contrast technique and OM, as shown in Figs. 3 and 4. In the figures, the bright and dark regions represent the aluminum alloy and the mild steel, respectively.

For the both FSW parameter combinations, the results of the OM of the cross sections reveal that that the shoulder influenced area (SIF) is limited to top part of the SZ, whereas the bottom part of the SZ was influenced by the pin (the pin influenced area: PIF). In the SZ, the material flow took place along the horizontal and vertical paths, as indicated by the blue arrows. Material flow in the SIF indicates the transfer of aluminum alloy from the retreating side to the advancing side along the horizontal path. In the PIF, the material flow is nearly opposite. The steel from the advancing side

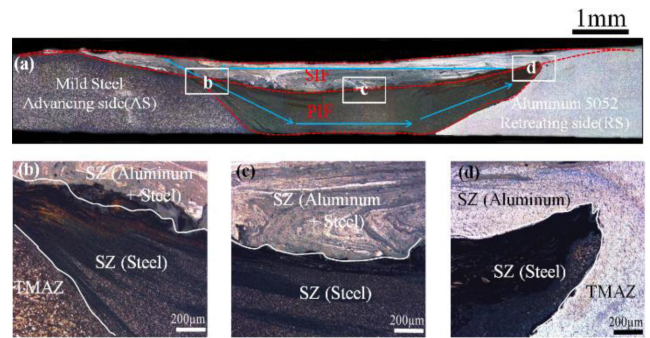


Fig. 3 (a) Material flow path and weld cross section for 800/75 combination of parameters (b), (c), and (d) are magnified zones in advancing, central and retreating side as marked in the weld cross section

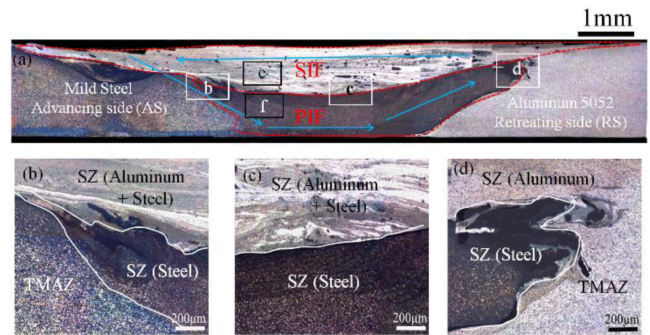


Fig. 4 (a) Material flow path and weld cross section for 1000/75 combination of parameters (b), (c), and (d) are magnified zones in advancing, central and retreating side as marked in the weld cross section

was extruded and penetrated into the aluminum alloy in the retreating side, exhibiting an elongated band structure in the bottom part of the SZ. Intermixing of the steel and the aluminum alloy with formation of lamellar structures inside the SIF is clearly observed, as shown in the magnified views in Figs. 3(b)-3(d) and Figs. 4(b)-4(d). The results of EDS elemental scan for the SIF and PIF with 1000 rpm / 75 mm/min (Fig. 5) confirm the observation. This suggests that a certain quantity of the steel extruded to the retreating side was again re-transported to the advancing side. While the material flow paths for the both FSW parameter combinations are generally similar, the higher heat input by the parameter combination with 1000 rpm naturally plasticize the materials to a higher extent. As a result, a larger amount of aluminum alloy was transferred from the retreating side and a thicker SIF was formed in the SZ for the parameter combination with 1000 rpm.

The Vickers hardness distributions were measured along the three different lines on the cross sections of FSW joint, as depicted in Figs. 6(a)-6(d). For the both parameter combinations, the

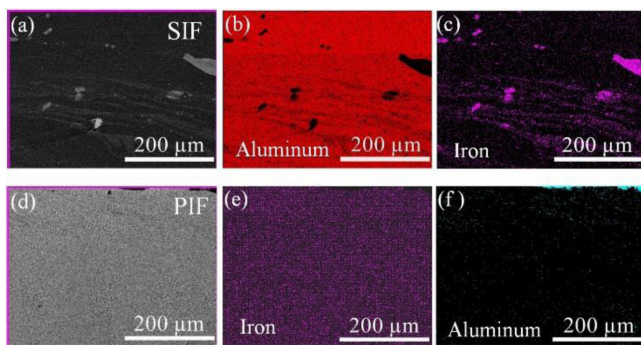


Fig. 5 (a) SEM image of the center of SIF with 1000 rpm / 75 mm/min and the results of EDS elemental scan of the region shown in (a): (b) aluminum and (c) iron; (d) SEM image of the center of PIF with 1000 rpm / 75 mm/min and the results of EDS elemental scan of the region shown in (d): (e) aluminum and (f) iron

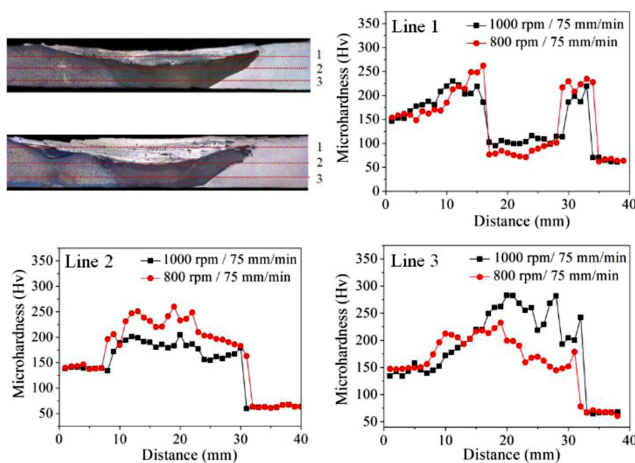


Fig. 6 Vickers microhardness results of (a) weld cross sections for 800/75 and 1000/75 parameters along line (b) 1, (c) 2, and (d) 3

hardness distribution along the line 1 passing the SIF shows that the hardness of the SIF is close to that of the base aluminum alloy, which is reasonable since the SIF is mostly composed of the aluminum alloy. The slight higher hardness is probably due to the intermixing of the steel from the advancing side in the SIF. It is interesting to see the hardness distributions in the PIF (lines 2 and 3) are significantly higher even than that of the base steel. Actually, it is very likely that intermetallic particles, which affect the mechanical properties significantly,<sup>18</sup> were formed in the SZ. Also the occurrence of dynamic recrystallization in the SZ is quite likely during FSW. Therefore, further understanding of the mechanical properties of the FSW joint of the steel and the aluminum alloy should be accompanied by thorough microstructural analysis. Additional microstructural analysis is beyond the scope of this technical report and will be conducted as a future work.

#### 4. Conclusion

Butt joining of the mild steel and the aluminum 5052-O alloy was successfully conducted by FSW with offset of the tool pin towards the steel sheet at the advancing side. The result of OM shows that the SZ is composed of two distinctively recognized regions, the SIF and the PIF. Due to the structure of the SZ, the hardness distributions on the cross section became quite different from each other, depending of the relative position of the measurement line. It is very likely that intermetallic particle formation and dynamic recrystallization occurred in the SZ during joining. A detailed microstructural analysis will be conducted as a future work. It also needs to be noted that the effect of other FSW parameters including tool travel speed and tool geometry could affect the joint property.

#### ACKNOWLEDGEMENT

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